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# **Room-temperature high performance H<sub>2</sub>S sensor based on porous CuO nanosheets prepared by hydrothermal method**

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## **Abstract**

Porous CuO nanosheets were prepared on alumina tubes using a facile hydrothermal method, and their morphology, microstructure and gas sensing properties were investigated. The monoclinic CuO nanosheets had an average thickness of 62.5 nm and were embedded with numerous holes with diameters ranging 5 nm to 17 nm. The porous CuO nanosheets were used to fabricate gas sensors to detect hydrogen sulfide (H<sub>2</sub>S) operated at room temperature. The sensor showed a good response sensitivity of 1.25 with the respond/recovery time of 234 s and 76 s, respectively, when tested with the H<sub>2</sub>S concentrations as low as 10 ppb. It also showed a remarkably high selectivity to the H<sub>2</sub>S, but only minor responses to other gases such as SO<sub>2</sub>, NO, NO<sub>2</sub>, H<sub>2</sub>, CO and C<sub>2</sub>H<sub>5</sub>OH. The working principle of the porous CuO nanosheets based sensor to detect the H<sub>2</sub>S was identified to be the phase transition from semiconducting CuO to a metallic conducting CuS.

Keywords: CuO; Nano-sheet; Hydrogen sulfide; Hydrothermal method; Gas Sensor

## 1. Introduction

H<sub>2</sub>S is one of the most common toxic contaminants, which are frequently utilized in various fields, including oil, gas, waste treatment and paper industries.<sup>1,2</sup> It is also commonly found or generated from sewage, rubbish dumps as well as many routine chemical production processes. Even in the presence of trace amounts, the H<sub>2</sub>S gas is extremely toxic to many organisms, such as human respiratory and nerve system. Generally, it is recommended that the acceptable ambient levels of the H<sub>2</sub>S for a healthy condition are in the range of 20-100 ppb.<sup>3</sup> Therefore, in the views of environmental protection, safety and health conditions of human being, it is urgently required to develop cheap, efficient, highly sensitive and mass-produced H<sub>2</sub>S sensors working at room temperature, also with other advantages of excellent selectivity and reliability even as low as ppb concentration. Various H<sub>2</sub>S gas sensors fabricated using different types of semiconductor oxides have been investigated, including In<sub>2</sub>O<sub>3</sub>,<sup>4,5</sup> ZnO,<sup>6</sup> SnO<sub>2</sub>,<sup>7</sup> WO<sub>3</sub>,<sup>8</sup> Fe<sub>2</sub>O<sub>3</sub> and CuO.<sup>9-14</sup> Among them, CuO based nano-materials have received significant attention recently due to its excellent sensing performance.

Generally, chemical, physical, mechanical and optical properties of the nanomaterials are strongly dependent on their nanostructure and morphology. There are lots of reports recently on the synthesis of various types of the CuO nanostructures, which include nanoparticles,<sup>15</sup> nanoneedle,<sup>16</sup> nanowires,<sup>17</sup> nanoflowers,<sup>18,19</sup> nanotube,<sup>20</sup> nanorods,<sup>21</sup> nanoleaves and nanosheets.<sup>22-29</sup> Among these, the 2D sheet-like CuO nanostructures have received considerable attention because of their high anisotropy and nano-scale

thickness. The nano-porous structures of CuO sheets could allow a fast and efficient gas adsorption on their surfaces, thus the response times of gas sensor will be significantly decreased. However, as far as we have known, there are few reports using these CuO sheet-like nano-porous structures for gas sensing applications.

The sensing mechanism of a commonly used semiconductor gas sensor is based on the reactions of the absorbed target molecules with the metal oxide materials. For the electrical resistance based metal oxide H<sub>2</sub>S sensors, these reactions will occur between the H<sub>2</sub>S molecule and the oxygen ions on the surface of the metal oxides, thus generating free electrons and thus resulting in changes in the electrical resistance of the metal oxides. However, the similar reaction on the CuO materials and the associated changes in the resistance are also commonly observed in absorption of other types of reducing gases, including CO,<sup>11,30</sup> alcohol,<sup>31,32</sup> methane and ammonia gases.<sup>33,34</sup> Therefore, the specific H<sub>2</sub>S sensors with a good selectivity should be explored.

In this paper, using a hydrothermal preparation process, porous CuO nanosheets were prepared on alumina tubes, on which, the resistance based H<sub>2</sub>S gas sensors were fabricated, and their sensing performance was systematically studied.

## 2. Experiment

### 2.1 Preparation of porous CuO nanosheets

Copper chloride ( $\text{CuCl}_2 \cdot \text{H}_2\text{O}$ ), NaOH and sodium dodecyl benzene sulfonate ( $\text{C}_{18}\text{H}_{29}\text{NaO}_3\text{S}$ ) of analytical grades (Sinopharm Co. Ltd., China) and  $\text{H}_2\text{O}$  (with a resistivity reading of  $18.0 \text{ M}\Omega \cdot \text{cm}$ ) were used. In a typical synthesis process,  $\text{CuCl}_2 \cdot 2\text{H}_2\text{O}$  of 1.70 g were dissolved into 25 ml distilled water under a continuous stirring at room temperature to form a 0.4 M  $\text{CuCl}_2 \cdot 2\text{H}_2\text{O}$  homogeneous solution. Subsequently,  $\text{C}_{18}\text{H}_{29}\text{NaO}_3\text{S}$  of 3.48 g was mixed into the  $\text{CuCl}_2 \cdot 2\text{H}_2\text{O}$  solution under a continuous stirring. NaOH (4 mol/L) with 15 ml volume was gradually dripped into the prepared solutions within 10 minutes to obtain a blue-colour solution. The obtained solution was placed inside a 50 ml Teflon-lined stainless autoclave, and tubes of alumina (1.5 mm of outer diameter and 4 mm long) were placed vertically on the bottom of the autoclave. The chemical reactions in the autoclave took about 24 hours with a constant temperature oven of  $120 \text{ }^\circ\text{C}$ . After the reaction, it was naturally cooled down to room temperature ( $\sim 25 \text{ }^\circ\text{C}$ ). The alumina tubes were found to be covered with a layer of black precipitates of CuO materials. These alumina tubes were flushed with the distilled water for 3 times, and then washed with alcohol for 3 times. Finally, the alumina tubes with CuO materials were dried at  $55 \text{ }^\circ\text{C}$  for 6 hours and annealed for 2 hours in air at  $600 \text{ }^\circ\text{C}$ .

### 2.2 Characterization of CuO samples

The XRD spectrum of the porous CuO nanosheets was measured by a D/MAX2500 diffractometer with a copper target of  $1.5406 \text{ \AA}$  of wavelength, operated with

voltage/current of 40 kV/30 mA. Scanning electron microscope (SEM, Inspect F50) was used to investigate the morphologies of the CuO nanosheets. The crystallographic features were also analyzed using a scanning transmission electron microscope (STEM, JEOL 3100R5) operated using a 300 kV Cold field emission gun (STEM). Surface areas of the CuO nano-sheets were characterized based on the Brunauer–Emmett–Teller (BET) method. The measurement was operated using an equipment (Tristar 3000, Micromeritics) for detecting the N<sub>2</sub> adsorption on the nanosheet surfaces at a temperature of 77 K. Chemical binding analysis was performed by an X-ray photoelectron spectrograph (XPS, KratosAxis-Ultra DLD), operated using a monochromatic Al K $\alpha$  source with a source energy of 1486.6 eV. UV–vis spectroscopy was recorded using a UV-2101 spectrophotometer (Shimadzu Corporation, Japan).

### 2.3 Manufacture and testing of gas sensor

The alumina tubes coated with the porous CuO nanosheets were used to directly fabricate the sensors. The surfaces of the alumina tubes were covered with a layer of the porous CuO nanosheets layer (as shown in the schematic illustration of Figure S1 in the Supporting Information).<sup>9</sup> At opposite ends of the alumina tube, there was a Pt wire connected to gold electrode. The electric current of the sensor was recorded using a Keithley 2400 source meter with an applied working voltage of  $V_s = 0.25$  V. In this paper, the definition of gas response (S) for the gas sensor is following the ideas reported in literature<sup>9</sup>:  $S = R_a/R_g$ , in which  $R_g$  and  $R_a$  are the resistance of CuO nanosheets layer measured in the H<sub>2</sub>S and air, respectively.<sup>9</sup> All the testing gases were

commercially ones. In a standard testing process, the gas sensor was placed in a chamber of two liters and a fixed relative humidity of 30%. During the gas sensing test, the concentrations of the gas were adjusted based on the injected volumes of the H<sub>2</sub>S.

### 3. Results and discussion

#### 3.1 Structural characterization

The SEM images of the CuO samples are shown in Figure 1. Clearly, the CuO nanosheet structures can be identified, which have the average widths and lengths of about 0.5  $\mu\text{m}$  and 1.2  $\mu\text{m}$ , respectively. The average thickness of the CuO nanosheets is 62.5 nm. The STEM image shown in Figure 2a confirms the sheet-like CuO structures. From the STEM dark field image in Figure 2b, there are many pores with diameters ranging from 5 nm to 17 nm inside these nanosheets (see the areas with red circles). The porous structure is obviously different from the literature for those leaves-like or sheet structures of CuO.<sup>25-39</sup> The surface area of these CuO nanosheets obtained based on the BET method is 10.03  $\text{m}^2\cdot\text{g}^{-1}$ . These porous CuO nanosheets are efficient for the gas flowing in and out, thus enhancing their gas sensing performance. The crystalline structure in Figure 2c shows a well-defined crystal structure with lattice spacings of 0.231 and 0.196 nm, which were identified as the (200) and (20 $\bar{2}$ ) planes of CuO, respectively.

The XRD spectrum of the porous CuO samples is shown in Figure 3, revealing the monoclinic CuO structures (JCPDS No. 48-1548). The lattice constants are  $a = 0.4688$ ,  $b = 0.3423$  and  $c = 0.5132$  nm. No characteristic peaks from the other phases were



identified, meaning that the porous CuO nanosheets are composed of a single phase monoclinic CuO.

In the synthesis process of the porous CuO nanosheets, the ratio of  $\text{Cu}^{2+}$  and NaOH was 1:6. It was previously reported that, in the NaOH aqueous solution with a high concentration, the  $\text{Cu}^{2+}$  ions produced a square-planar complex  $[\text{Cu}(\text{OH})_4]^{2-}$  ions, rather than  $\text{Cu}(\text{OH})_2$  (as can be seen from chemical equation 1 below).<sup>35</sup>



During decompositions in the hydrothermal process, because of coordination self-assembly of  $[\text{Cu}(\text{OH})_4]^{2-}$  ions,  $\text{Cu}(\text{OH})_2$  is easily formed into 2D layered orthorhombic nanostructures (see chemical equation 2).<sup>36</sup> Therefore, once the nuclei was formed on the surface of alumina tube, the  $\text{Cu}(\text{OH})_2$  nanosheets could start to grow. It is likely that some sodium dodecyl benzene sulfonate ligands were trapped in these nanosheets. The  $\text{Cu}(\text{OH})_2$  was then transformed into CuO structures in the post-hydrothermal process which is under high temperature and pressure (see chemical equation 3). At the same time, the trapped sodium dodecyl benzene sulfonate was released and nano-pores were formed in the CuO nanosheets.

The porous CuO nanosheets showed a broad absorption from 200 nm to 800 nm (as shown in the UV-vis absorbance spectrum of Figure S2a in the Supporting Information). The band gap energy can be obtained by the standard Tauc's relationship.<sup>37</sup> Based on this classical Tauc approach, the  $E_g$  value of the porous CuO nanosheets of 3.08 eV can be obtained by extrapolating the value at  $\alpha=0$  (as shown in

the  $(\alpha h\nu)^2$  vs  $h\nu$  curve of Figure S2b in the Supporting Information). This value is significantly larger than the band gap energy values of the bulk CuO crystals of 1.85 eV.<sup>37</sup> The larger  $E_g$  values of various CuO nanostructures have been reported, such as 3.48 eV for ultra-long CuO nanowires,<sup>38</sup> 3.55 eV for CuO nanoplates,<sup>39</sup> and 3.02 eV for CuO nanoplatelets.<sup>40</sup> It has been reported that the specific sites of molecular adsorption were remarkably influenced by the band gap energy, and thus could remarkably influence the gas sensitive property.<sup>37,38</sup>

### 3.2 Sensing performance

Figure 4 shows the response/recovery curves of the gas sensor based on CuO nanosheets to H<sub>2</sub>S from 10 ppb to 60 ppm at room temperature. The porous CuO nanosheets sensor provides a stable baseline in the dry air condition before the sensing measurement at different concentrations. Upon the H<sub>2</sub>S injection, a positive sensor response was obtained, meaning that resistance was decreased drastically. After the H<sub>2</sub>S gas was pumped away, the signal was returned back to its initial stable baseline. The sensitivity values of the porous CuO nanosheets sensor to various concentrations of the H<sub>2</sub>S were calculated, and the results are shown in Figure 5a. The inset of Figure 5a shows the sensitivity values to sub-ppm concentrations of H<sub>2</sub>S. The sensitivity of the sensor increases with the increasing concentrations of the H<sub>2</sub>S. Even with the H<sub>2</sub>S concentration as low as 10 ppb, the sensor still has good response sensitivity value about 1.25 at room temperature as shown in Figure 4a and Figure 5a.

Most previously published papers reported that the H<sub>2</sub>S sensing could be successfully performed at a relatively high temperature. For example, Steinhauer et

al<sup>11</sup> reported the detection of 10 ppb H<sub>2</sub>S using a CuO nanowire based sensor at 325 °C, and Zhang et al<sup>14</sup> reported the detection of 30 ppb H<sub>2</sub>S using CuO nanosheet based sensor at 240 °C. Although Ramgir et al reported a room temperature operated sensor using CuO thin films, the detection limit to H<sub>2</sub>S was higher than 100 ppb.<sup>12</sup> Therefore, we can confirm that at room temperature, our sensor based on the porous CuO nanosheets in this study has an excellent property for detecting hydrogen sulfide at lower concentrations.

Figure 5b shows the response/recovery times of the porous CuO nanosheets based sensor. The definition of response/recovery times was based on the report of the literature.<sup>9</sup> There was an obvious change of the response/recovery times based on the results shown in Figure 5b. The response times were in the range from 41 s to 606 s when the H<sub>2</sub>S concentrations were changed from 10 ppb to 60 ppm. However, when the H<sub>2</sub>S gas concentration was higher than 1 ppm, it was less than 90 s. The recovery time is in the range from 17 s to 1173 s, but it was less than 76 s when the concentration of H<sub>2</sub>S was less than 0.04 ppm. It can be concluded that the huge amount of nanopores in the nanosheets (as shown in Figure 2b) are favorite for both absorption/desorption of H<sub>2</sub>S gas, leading to the shorter response/recovery times.

Besides the high sensitivity, the reproducibility of the gas sensor is another important performance indicator. Figure 6a shows the reproducibility testing results of the porous CuO nanosheets based sensor, which was successively exposed to 200 ppb H<sub>2</sub>S for five times at the room temperature. Clearly a good reproducibility was obtained. During the repeated absorption/desorption of the H<sub>2</sub>S, the dynamic curves

for five times are almost identical. It shows a stable response curve with a maximum response sensitivity about 5.01 to 200 ppb H<sub>2</sub>S.

Long-term stability is also a critical parameter for the application of gas sensor. The measurement readings of the porous CuO nanosheets based H<sub>2</sub>S sensor were recorded for a month, and the result are plotted in Figure 6b. The response deviation when the device was exposed in 200 ppb H<sub>2</sub>S is lower than 5% after long-term testing for a month, indicating that the CuO nanosheet sensor has a good long-term stability.

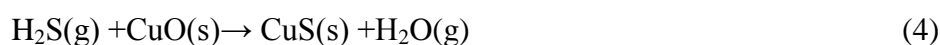
The selectivity of a gas sensor is another critical parameter. Figure 7 shows the response curves of the sensor exposed to various gases (SO<sub>2</sub>, NO, NO<sub>2</sub>, H<sub>2</sub>, CO, C<sub>2</sub>H<sub>5</sub>OH and NH<sub>3</sub>) at the same gas concentration of 40 ppm compared with the H<sub>2</sub>S of 0.2 ppm at room temperature. It is obvious that, compared with the remarkable responses to the 0.2 ppm H<sub>2</sub>S, there was no responses of the gas sensor to 40 ppm SO<sub>2</sub>, NO, NO<sub>2</sub>, CO, H<sub>2</sub> and C<sub>2</sub>H<sub>5</sub>OH. Only one exception is the 40 ppm of NH<sub>3</sub>, with a low response sensitivity of 1.42. However, it is clear that the response sensitivity of 0.2 ppm H<sub>2</sub>S was 3.5 times higher than that to the 40 ppm NH<sub>3</sub>. The results indicated that there is a good selectivity of the porous CuO nanosheets H<sub>2</sub>S sensor compared with the previously reported results.<sup>11,33,41</sup>

### 3.3 Gas-sensing mechanism

CuO is a p-type semiconductor, and the charge carriers are positive holes. The adsorption and desorption of the H<sub>2</sub>S on the surface of the CuO will result in the changes of electrical resistance. According to the mechanism of oxidation of H<sub>2</sub>S by the adsorbed oxygen ions, the resistance of the CuO will be increased.<sup>42,43</sup> However,

from the real-time response shown in Figure 8, it is worth mentioning that the resistance of the sensor decreased quickly upon the H<sub>2</sub>S injection, which is opposite to the commonly accepted mechanisms for the p-type semiconductor gas sensors. It is apparent that there is a totally different sensing mechanism of the porous CuO nanosheets with the H<sub>2</sub>S. In this paper, we proposed that the working principle of sensor should be the transformation from CuO to CuS, the latter of which has a good metallic-like conductivity.

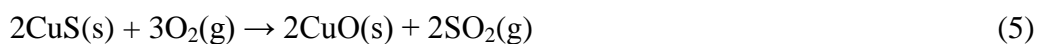
When the H<sub>2</sub>S molecules are absorbed on the CuO surface, it will react with the CuO directly, based the chemical reaction 4 as shown below:<sup>43</sup>



Firstly, the Cu<sub>x</sub>S (x<1) layer is formed on the CuO surface, and then it converts to CuS layer, which covers the surface of CuO sheets. The S<sup>2-</sup> ions continue to percolate inside the bulk CuO and form Cu<sub>x</sub>S percolation regime in the sub-layer, as shown in Figure 9a. Formation of the CuS on the CuO has been proven by the XPS analysis results.<sup>43,44</sup> CuS, a metallic-like conductor, will increase the connectivity between the neighboring CuO sheets, leading to a decrease of the resistance of the CuO nanosheets film. The porous CuO nanosheets are convenient for the adsorption of the H<sub>2</sub>S gas molecules, thus facilitating the transformation of CuS. Because of existence of the porous structures of the sheet-like CuO nanostructures, the reaction of CuS formation is very fast. It was 336 s to achieve a reaction equilibrium for the 200 ppb H<sub>2</sub>S gas at room temperature as shown in Figure 8.

After the H<sub>2</sub>S gas was replaced with the dry air, recovery of the measured response

was initiated immediately. Based on a schematic illustration of the recovery pathway as shown in Figure 9b, the CuS will transform back into CuO in air based on the chemical reaction 5 as shown below:<sup>43</sup>



The porous structures of the CuO nanosheets are convenient for the molecular diffusion/desorption. Therefore, the reading of the sensor can be quickly recovered at room temperature as shown in Figure 8, revealing its reliability to the applications.

XPS results confirmed the formation of CuS. Because it is quite difficult to characterize the H<sub>2</sub>S sensing process using XPS, here, the XPS analysis was only used to prove that the CuS has formed on the surface of CuO powder. The testing sample used in the XPS analysis was CuO powder. The CuO powder was firstly placed in a chamber with the H<sub>2</sub>S gas at room temperature, then it was taken out and measured using XPS. The CuO powder without exposed to the H<sub>2</sub>S gas was also characterized using the XPS for a comparison. Before and after exposure in H<sub>2</sub>S gas, the XPS spectra of Cu 2p<sub>3/2</sub> and S 2p of the CuO nanosheets are shown in Figure 10, respectively. Before exposure in H<sub>2</sub>S gas, the Cu 2p<sub>3/2</sub> spectrum shows a main peak at 933.9 eV along with its satellite peaks, which are all attributed to those of the CuO.<sup>18,45</sup> However, for the CuO sample which had been exposed in H<sub>2</sub>S gas, the main peak of the Cu 2p<sub>3/2</sub> shows a broad asymmetric curve which was deconvoluted into two peaks at 932.6 and 930.8 eV, attributing to CuO and CuS, respectively.<sup>45,46</sup> For the S 2p spectra, the peaks at 162.3 and 163.4 eV could be identified (Figure 10b) after exposure to H<sub>2</sub>S, which are attributed to S 2p<sub>3/2</sub> and S 2p<sub>1/2</sub> states respectively.

This result clearly confirms the formation of CuS.<sup>46,47</sup> When it was exposed to air again, the XPS spectra were found to be almost the same as those before exposure to the H<sub>2</sub>S, indicating the transformation from the CuS to CuO.

As shown in Figure 7, when the sensor was exposed into ammonia, the resistance of the sensor was also reduced, although compared with the results from the H<sub>2</sub>S, its response sensitivity is quite low. The sensing mechanism to the NH<sub>3</sub> for the CuO sensor could be explained based on proton conductivity of NH<sub>4</sub><sup>+</sup>.<sup>48</sup> The H<sub>2</sub>O molecules were absorbed on porous CuO nanostructures at room temperature. The absorbed NH<sub>3</sub> molecules could react with the H<sub>2</sub>O molecules, based on the following chemical reaction:



Therefore, the resistance of gas sensor was decreased due to the proton conductivity of NH<sub>4</sub><sup>+</sup>.<sup>48</sup>

#### **4. Conclusions**

In this paper, we presented the fabrication and characterization of porous CuO nanosheets on the alumina tubes. And the fabricated H<sub>2</sub>S sensor showed an excellent sensing performance. The CuO nanosheets had an average thickness of 62.5 nm. Detailed structural analysis confirmed that the synthesized CuO nanosheets had monoclinic CuO structures, and there were many holes with a diameter ranging from 5nm to 17 nm inside the nanosheets. The working principle of sensor was attributed to the transformation of semiconducting p-type CuO to metallic CuS. The porous CuO nanosheets based H<sub>2</sub>S sensor showed excellent gas sensitive performance and

remarkably good selectivity, which showed little responses to SO<sub>2</sub>, NO, NO<sub>2</sub>, H<sub>2</sub>, CO and C<sub>2</sub>H<sub>5</sub>OH at room temperature, and a relatively weak response to NH<sub>3</sub>. Therefore, the porous CuO nanosheets based sensor prepared in this study can be efficiently used for high-performance H<sub>2</sub>S gas sensor.

## **ASSOCIATED CONTENT**

Supporting Information available:

Measurement schematic illustration of gas sensor based on the porous CuO nanosheets sensing material.

UV–vis absorptions spectrum and (b) the corresponding plots of  $(\alpha h\nu)^2$  vs  $h\nu$  curve of the porous CuO nanosheets.

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### **Notes**

The authors declare no competing financial interest.

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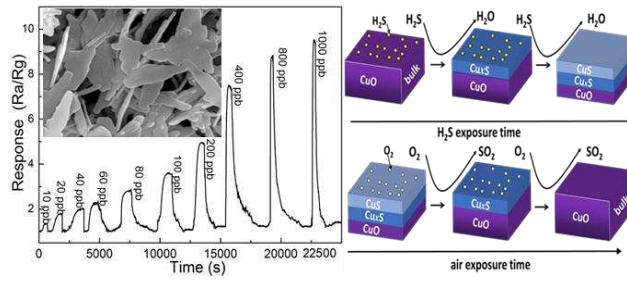
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# The Table of Contents (TOC) Graphic





## Figures

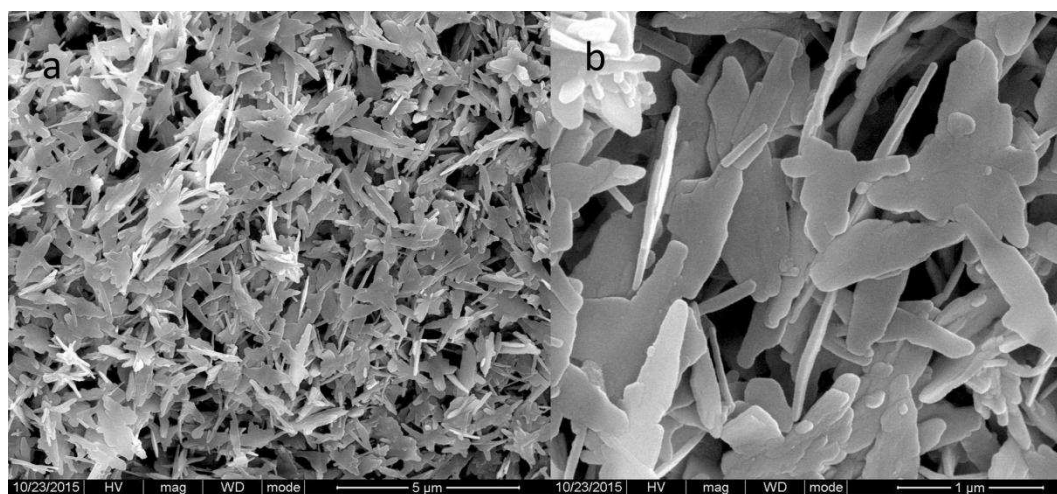


Figure 1. SEM images with different magnifications of the CuO sample (a) Low magnification; (b) High magnification.

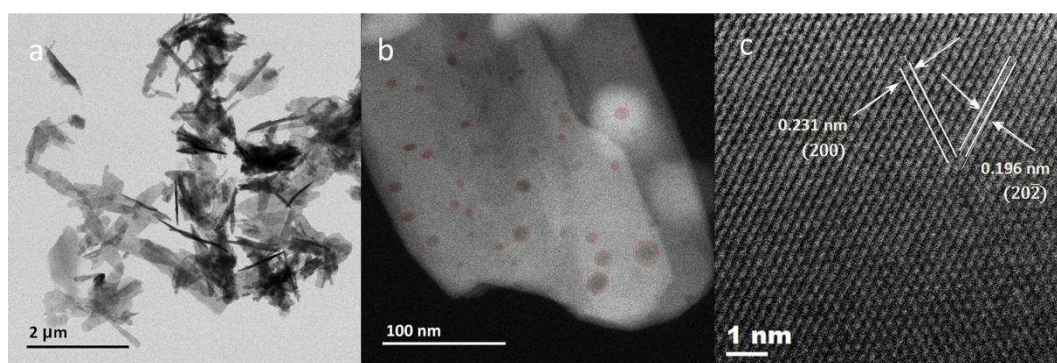


Figure 2. (a) STEM image of low magnification bright field; (b) STEM image of dark field image; (c) the crystalline framework of the CuO samples.

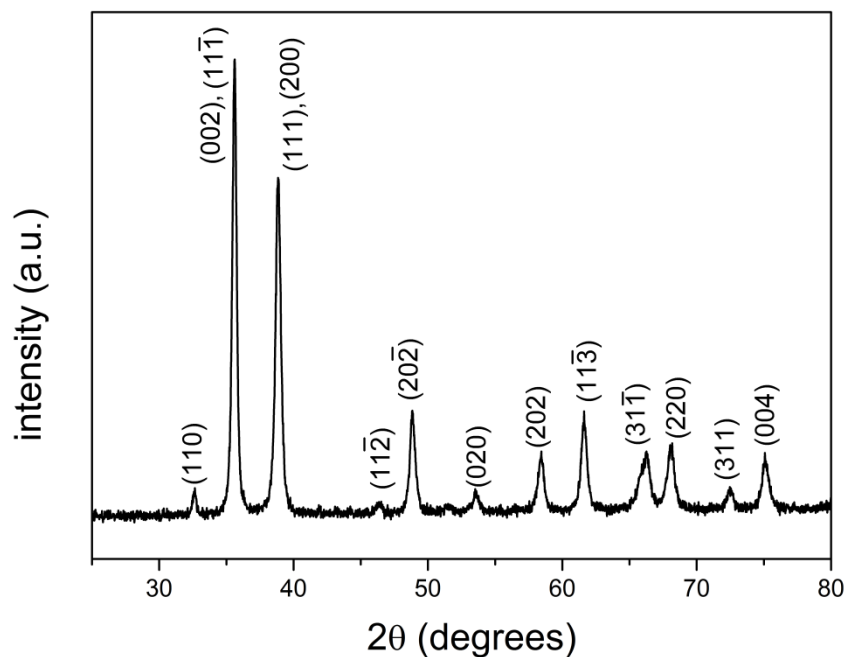


Figure 3. XRD pattern of the porous CuO nanosheets.

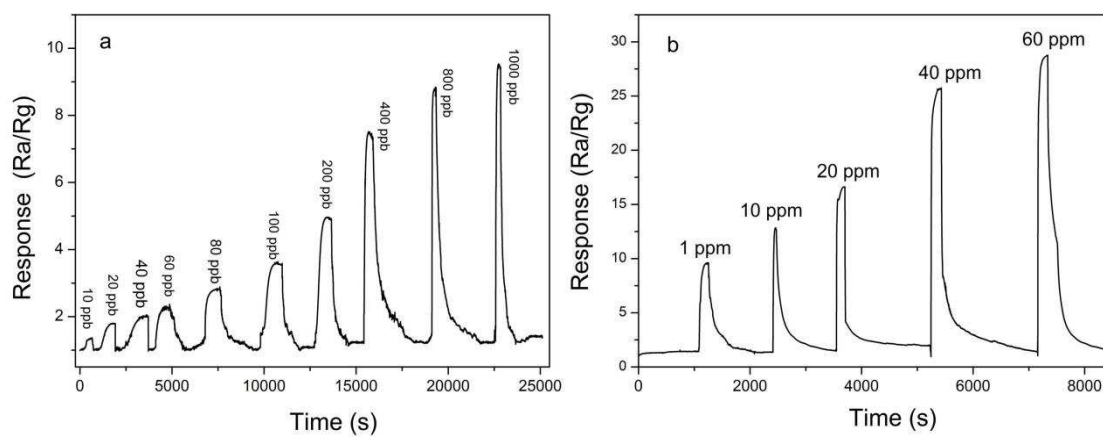


Figure 4. Response and recovery curves of the porous CuO nanosheets based gas sensor to  $H_2S$  with the concentration (a) from 10 ppb to 1 ppm and (b) from 1 ppm to 60 ppm at room temperature.

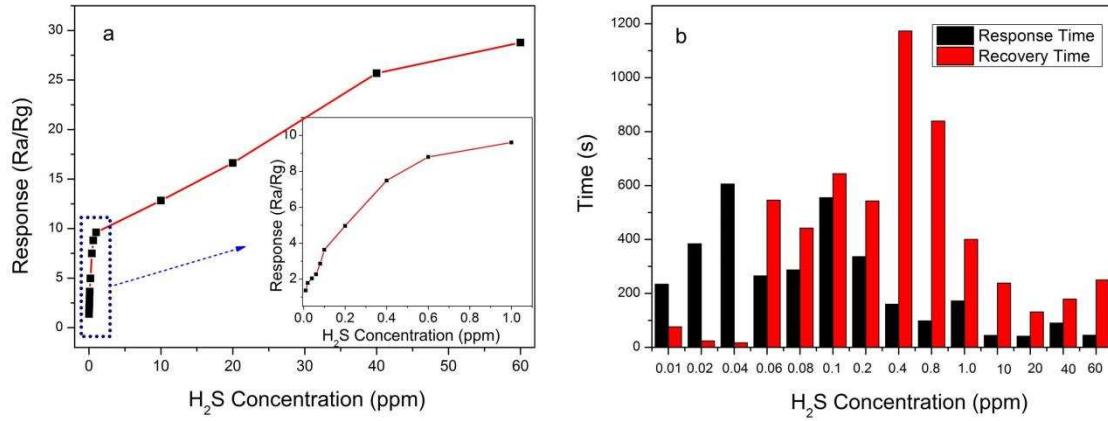


Figure 5. (a) Variation of the sensitivity and (b) Response time and recovery time of the porous CuO nanosheets based gas sensor at different H<sub>2</sub>S gas concentration from 10 ppb to 60 ppm.

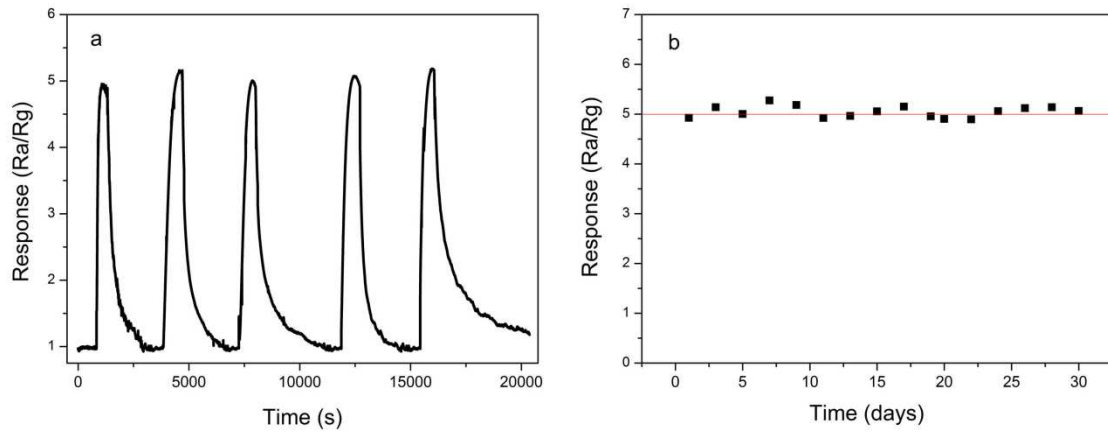


Figure 6. (a) Reproducibility and (b) Long-term stability of the porous CuO nanosheets based gas sensor exposed to 200 ppb H<sub>2</sub>S gas for five times.

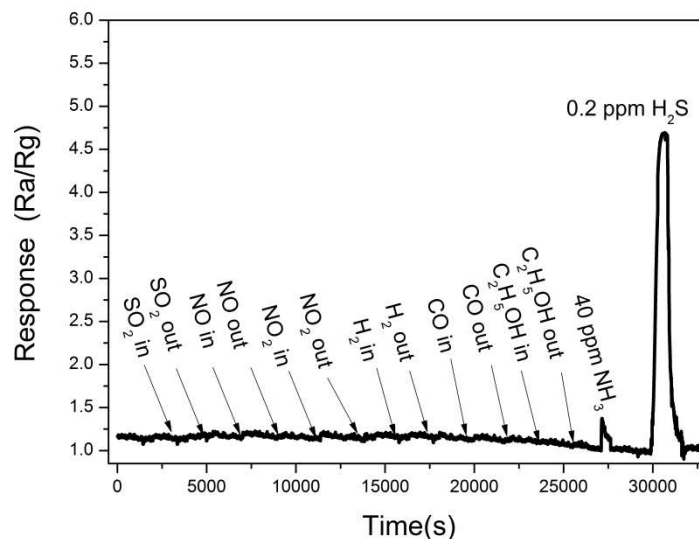


Figure 7. Response and recovery curves of the porous CuO nanosheets based gas sensors to difference gases ( $\text{SO}_2$ , NO,  $\text{NO}_2$ ,  $\text{H}_2$ , CO,  $\text{C}_2\text{H}_5\text{OH}$ ,  $\text{NH}_3$ ) of 40 ppm and 0.2 ppm  $\text{H}_2\text{S}$  at room temperature.

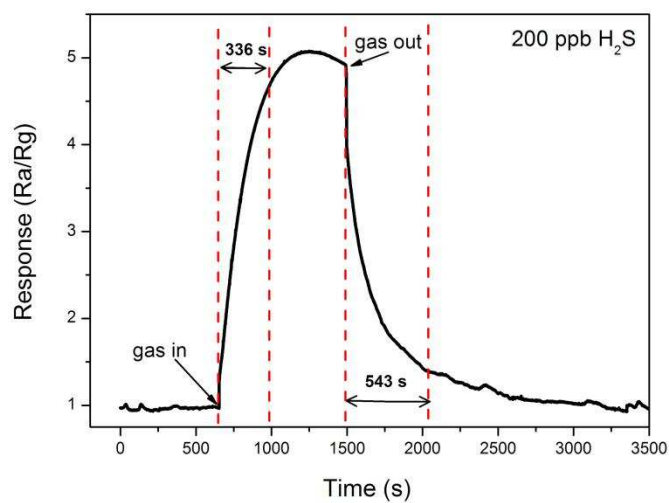


Figure 8. Real-time gas sensing curve of the sensor based on the porous CuO nanosheets to 200 ppb  $\text{H}_2\text{S}$  gas.

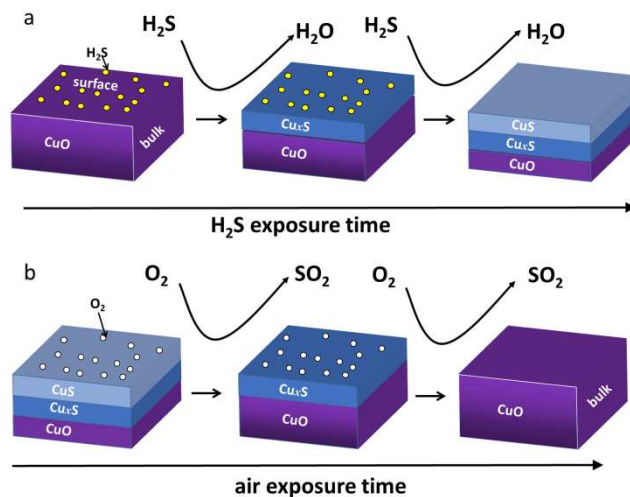


Figure 9. Schematic illustration of (a) reaction pathway in  $\text{H}_2\text{S}$  and (b) recovery pathway in air of CuO nanosheet.

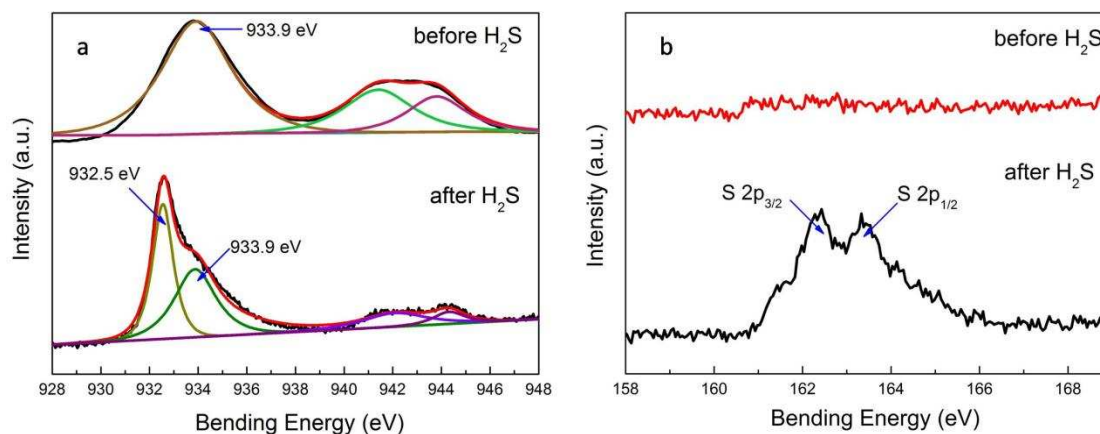


Figure 10. (a)  $\text{Cu } 2p_{3/2}$  and (b)  $\text{S } 2p$  XPS spectra of the CuO samples before and after exposure to  $\text{H}_2\text{S}$ .